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# Benchmarking van der Waals Density Functionals with Experimental Data: Potential Energy Curves for H<sub>2</sub> Molecules on Cu(111), (100), and (110) Surfaces

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Detailed physisorption data from experiment for the H<sub>2</sub> molecule on low-index Cu surfaces challenge theory. Recently, density-functional theory (DFT) has been developed to account for nonlocal correlation effects, including van der Waals (dispersion) forces. We show that the functional vdW-DF2 gives a potential-energy curve, potential-well energy levels, and difference in lateral corrugation promisingly close to the results obtained by resonant elastic backscattering-diffraction experiments. The backscattering barrier is sensitive to the choice of exchange-functional approximation. Further, the DFT-D3 and TS-vdW corrections to traditional DFT formulations are also benchmarked, and deviations are analyzed.

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## I. INTRODUCTION

The van der Waals (vdW) or dispersion interactions play important roles in defining structure, stability, and function for molecules and materials. Our understanding of chemistry, biology, solid state physics, and materials science benefits greatly from density-functional theory (DFT). This in-principle exact theory for stability and structure of electron systems [1] is computationally feasible also for complex and extended systems. However, in practice, approximations have to be made to describe exchange and correlation (XC) of the participating electrons [2]. This work aims at benchmarking some XC descriptions of nonlocal correlations that describe vdW interactions [3–5].

To boil down the intricate electronic dynamics behind the vdW interaction into a density functional  $F[n]$  is a formidable task.  $F[n]$  should depend only on the electron density  $n(\mathbf{r})$  and do that in a proper and generally applicable way. This means that it should obey fundamental physical laws, like charge conservation and time invariance, and have a physically sound account of system and interactions. The vdW-DF method [3, 6] has such ambitions. There are other vdW-corrected DFT schemes, specifically applied to physisorption of atoms and small molecules to metal surfaces, including those correcting traditional DFT calculations with pairwise vdW potentials, like the DFT-D [4] and TS-vdW [5] methods, and those that use maximally localized Wannier functions [7, 8]. Further, first-principles electron-structure calculations are made efficient but still carry much higher com-

putational costs than DFT. An example is the random-phase approximation (RPA) to the correlation energy used as a suitable complement to the exact exchange energy [9–11].

The results obtained from particular XC functionals and other vdW descriptions can be assessed by comparing with other accurate electron-structure theories like those presented in Refs. 12, 13, or with experiments. Typically one or two measurable quantities are available, like in Ref. 14. In Ref. 3 some of us stressed the importance of exploiting extended and accurate experimental data sets when these are available. Here, we extend this comparison by considering several facets of Cu.

Surface physics has a long and successful tradition of detailed and informative experiment-theory comparisons and offers possibilities also here. Extensive data sets are available for systems and conditions where the weak vdW forces can be reached and accurately mapped. A full physisorption potential and a detailed characterization thereof have been derived from versatile, accurate, and clearly interpretable measurements. In the physisorption regime, resonant elastic backscattering-diffraction experiments from low-index crystal faces provide a detailed quantitative knowledge. The actual data bank is rich and covers results, for instance for the whole shape of the physisorption potential, for the differences in corrugation across several facets, and for the energy levels in the potential well.

This paper compares state-of-the-art vdW descriptions of physisorption of H<sub>2</sub> and D<sub>2</sub> molecules on the low-index Cu surfaces with physisorption potentials constructed from selective-adsorption bound-state measurements. These data were analyzed in the early 90s in model systems [15–17]. In general terms, the measurements, calculations, and analysis underline the importance of building in the essential surface-physics into

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vdW functionals and other vdW accounts.

An earlier study of H<sub>2</sub> on the close-packed Cu(111) surface shows some spread in the results from different vdW accounts and that one of the tested XC functionals (vdW-DF2) compares promisingly with the experimental physisorption potential [18]. This motivates an extension of the study to the hydrogen molecule on other, more corrugated Cu surfaces. This paper is a significant extension of Ref. 18, addressing questions that were not resolved back in 1993 [15–17], for instance, trends with crystal face.

The outline, beyond this introduction, is as follows: First is a brief review of physisorption, in particular on metal surfaces, and a review of the traditional description. This is followed by a presentation of some DFTs with accounts of vdW forces, a presentation of the systems studied, and a review of the experimental benchmark sets. Next, calculated results for Potential-Energy Curves (PECs) and other physical quantities are presented, and the Paper is concluded with comparisons, analysis, and outlook for future functionals.

## II. PHYSISORPTION AND WEAK ADSORPTION

Chemically inert atoms and molecules adsorb physically on cold metal surfaces [17]. Characteristic desorption temperatures range from only a few K to tens of K, while adsorption energies range from a few meV to around 100 meV. These values may be determined from measurements of thermal desorption and isosteric heat of adsorption. For light adsorbates, like He and H<sub>2</sub>, gas-surface-scattering experiments provide a more direct and elegant method which involves the elastic backscattering with resonance structure. The bound-level sequences in the potential well can be measured with accuracy and in detail. Isotopes with widely different masses (<sup>3</sup>He, <sup>4</sup>He, H<sub>2</sub>, D<sub>2</sub>) are available. This permits a unique assignment of the levels and a determination of the well depth and ultimately a qualified test of model potentials [19].

The potential well is formed by the vdW attraction which arises from adsorbate-substrate electron correlation. At large distances from the surface the vdW attraction goes like  $V_{\text{vdW}}(z) = -C_{\text{vdW}}/z^3$ . Here  $z$  is the distance normal to the surface, measured from the center-of-mass of the particle to a surface reference plane close to the outermost layer of ion-cores in the solid, the so-called vdW plane. Near the surface the short-range repulsion, the “corrugated wall”, acts.

Specifically, we consider molecules that physisorb on metal surfaces where no significant change in the electronic configuration takes place upon adsorption. The weak coupling to electronic excitations [20] makes the adsorption largely electronically adiabatic. The energy transfer occurs through the phonon system of the solid lattice [21]. These conditions are expected to hold for hydrogen molecules on simple or noble metals.

Early atom- and molecule-diffraction studies typically were performed on single-crystal surfaces of ion-crystal surfaces. On metals, diffraction spots appear much weaker, which reflects the much weaker corrugation of close-packed metal surfaces [22], than on an ionic crystal, like LiF(100) [23].

In the traditional picture of physisorption, the interaction between an inert adparticle and a metal surface is approximated as a superposition of the long-ranged  $V_{\text{vdW}}$  and a short-range Pauli repulsion,  $V_R$ , a separation reminiscent of the Weeks-Chandler-Andersen approximation used early [24] to describe the role of repulsive forces in determining the equilibrium structure of simple liquids. The repulsive  $V_R$  is due to the overlap between wavefunction tails of the metal conduction electrons and the closed-shell electrons of the adparticle [17, 25, 26],

$$V_0(z) = V_R(z) + V_{\text{vdW}}(z). \quad (1)$$

Here approximately

$$V_R(z) = V'_R \exp(-\alpha z), \quad (2)$$

and

$$V_{\text{vdW}}(z) = -\frac{C_{\text{vdW}}}{(z - z_{\text{vdW}})^3} f(2k_c(z - z_{\text{vdW}})), \quad (3)$$

now with  $z$  measured from the “jellium” edge [27].  $V_0(z)$  is an effective potential. It arises as a lateral and adsorption-angle average of an underlying adsorption potential. We use  $V_1(z)$  to express the amplitude of the modulation around the average  $V_0(z)$ .

The repulsive potential  $V_R(z)$  has a prefactor  $V'_R$  that can be determined from the shifts of the metal one-electron energies caused by the adparticle. It can also be calculated by, for example perturbation theory in a pseudo-potential description of the adparticle and a jellium-model representation of the metal surface [15].

The strength of the asymptotic vdW attraction,  $C_{\text{vdW}}$ , and the reference-plane position,  $z_{\text{vdW}}$ , depend on the dielectric properties of the metal substrate and the adsorbate [28, 29]. The prefactor  $f(x)$  in the potential  $V_{\text{vdW}}(z)$  of Eq. (3) introduces a saturation of the attraction at atomic-scale separations. The function  $f(x)$  [ $f(x) = 1 - (1 + x + x^2/2) \exp(-x)$  in some accounts] lacks a rigorous prescription and thus includes some level of arbitrariness for  $V_{\text{vdW}}(z)$ . Experimental data provides a possible empirical solution to this dilemma.

The physisorption potential  $V_0(z)$  in Eq. (1) depends on the details of the surface electron structure both via the electron spill out ( $V_R$ ) and the spatial decay of polarization properties in the surface region ( $V_{\text{vdW}}$ ). Accordingly, there is a crystal-face dependence of  $V_0(z)$  for a given adparticle [17].

Figure 1 shows the electron density profiles calculated for the Cu(111), (100), and (110) surfaces with the method described below. They illustrate that the corrugations on these facets are small but differ, growing

in order  $(111) < (100) < (110)$ . For the scattering experiment, the density far out in the tails is particularly important.

That He-atom diffraction from dense metal surfaces is weak (compared to for instance ionic crystals) was early observed [22] and subsequently explained in terms of a simple tie between the scattering potential and the electron-density profile: The He-surface interaction energy  $E_{\text{He}}(r)$  can reasonably well be expressed as [30]

$$E_{\text{He}}(\mathbf{r}) \simeq E_{\text{He}}^{\text{hom}}(n_o(\mathbf{r})), \quad (4)$$

where  $E_{\text{He}}^{\text{hom}}(n)$  is the energy change on embedding a free He atom in a homogeneous electron gas of density  $n$ , and  $n_o(\mathbf{r})$  is the host electron density at point  $\mathbf{r}$ . On close-packed metal surfaces the electron distribution  $n_o(\mathbf{r})$  is smeared out almost uniformly along the surface [31], thus giving weak corrugation. The crude proposal (4) might be viewed as the precursor to the effective-medium theory [32].

The form (4) provides an interpretation of the mechanism by which the increasing density corrugation for  $(111) < (100) < (110)$  causes increasing amplitudes of modulation  $V_1(z)$  in the physisorption potentials. The min-to-max variation of the rotationally averaged, lateral periodic corrugation  $V_1(z)$  is modeled with an amplitude function [33], like in Eq. (2),  $V_1(z) = V'_1 \exp(-\beta z)$ . Here the exponent  $\beta$  is related to the exponent  $\alpha$  of  $V_0(z)$  via  $\beta = \alpha/2 + \sqrt{(\alpha/2)^2 + G_{10}^2}$ . The strength prefactor  $V'_1$  is adjusted so that the calculated intensities of the first-order  $\mathbf{G}_{10}$  diffraction beams agree with measured values.

The simple message of the experimental characterization in Figure 2(a) is that, at the optimal separation and out, the  $V_1$  corrugation terms are rather weak compared to the  $V_0$  averages. This observation confirms that the basic particle-surface interaction is predominantly one dimensional.

Figure 2(a) shows experiment-based PECs  $V_0(z)$  for physisorption of  $\text{H}_2$  on the Cu(111), Cu(100), and Cu(110) surfaces. The  $\text{H}_2$  molecules are trapped in states  $\epsilon_i$  that are quantized in the perpendicular direction but have an essentially free in-surface dynamics. The panel details the laterally (and rotationally) averaged potential  $V_0(z)$  that reflects the perpendicular quantization, i.e., the physisorption levels  $\epsilon_n$ . The experiment-based forms of  $V_0(z)$  [Fig. 2(a)] are obtained by adjusting the parameters<sup>1</sup> in the modeling framework [15–17], Eqs. (1)–(3), to accurately reproduce the set of  $\epsilon_n$  values. The

experiment-based PECs of Figure 2(a) are characterized by minima position (separations from the last atom plane), and depths given as follows: 3.52 Å and 29.0 meV for Cu(111), 3.26 Å and 31.3 meV for Cu(100), 2.97 Å and 32.1 meV for Cu(110). In Figure 2(a), however, the curves are shown with minima positions slightly translated so that the set of  $V_0(z)$  curves coincide at the classical turning point and thus facilitate an easy comparison.

The diffraction analysis of resonant back-scattering follows the reasoning: For light adsorbates, like He and  $\text{H}_2$ , in gas-surface-scattering experiments, the elastic backscattering has a resonance structure. This provides a direct and elegant method to characterize the PEC, as they give accurate and detailed measurements of bound-level energies  $\epsilon_n$  in the potential well. Isotopes with widely different masses, like  $^3\text{He}$ ,  $^4\text{He}$ ,  $\text{H}_2$ ,  $\text{D}_2$ , permit a unique assignment of the levels and a determination of the well depth and ultimately a qualified test of model potentials [19].

For a resonance associated with a diffraction that involves a surface reciprocal lattice vector  $\mathbf{G}$  there is a kinematical condition,

$$\epsilon_i = \epsilon_n + \frac{\hbar^2}{2m_p} (\mathbf{K}_i - \mathbf{G})^2, \quad (5)$$

where  $m_p$  is the particle mass and where  $\epsilon_i$  and  $\mathbf{K}_i$  are the energy and wavevector component parallel to the surface of the incident beam, respectively. At resonance, weak periodic lateral corrugations of the basic interaction induce large changes in the diffracted beam intensities. The narrow resonance is observed as features in the diffracted beam intensities upon variations in the experimental incidence conditions. The intrinsically sharp resonances in angular and energy space have line widths that depend on intermediate bound-state life-time. They are limited by elastic and phonon inelastic processes. Life-time broadening is only a fraction of a meV, substantially smaller than separations between the lower-lying levels (a few meV), allowing a number of physisorption levels  $\epsilon_n$  with a unique assignment to be sharply determined from Eq. (5).

$\text{H}_2$  is the only molecule for which a detailed mapping of the bound-level spectrum and the gas-surface interaction potential has been performed with resonance scattering measurements [15–17, 34–38]. The sequences here were obtained using nozzle beams of para- $\text{H}_2$  and normal- $\text{D}_2$ , that is, the beams are predominantly composed of  $j = 0$  molecules. Two isotopes  $\text{H}_2$  and  $\text{D}_2$  of widely different masses and with the different rotational populations

<sup>1</sup> The procedure takes off from the Le Roy analysis [15, 19] that gives an approximate determination of  $C_{\text{vdW}}$  (a complete direct specification of  $C_{\text{vdW}}$  would require measurements of even more shallow quantized physisorption levels) and of the depth of the physisorption well [16]. It also takes off from an approximate Zaremba-Kohn type [25, 26] specification of the repulsive wall. The location of the jellium edge [27] (relative to the position of the last atomic plane) is set at  $1.97 a_0$ , at  $1.71 a_0$ , and at  $1.21 a_0$  for the Cu(111), Cu(100) and for Cu(110) surfaces. The remaining set of parameters are split into two groups, those that

are constrained to be identical for all facets and those that are assumed to be facet specific. Fitting against the set of measured quantization levels  $\epsilon_n$  yields values  $C_{\text{vdW}} = 4740a_0^3$  meV,  $z_{\text{vdW}} = 0.563a_0$ , and  $k_c = 0.46a_0^{-1}$  for parameters in the first group, as well as the facet specific determinations,  $V'_R = 7480$  meV,  $V'_R = 5610$  meV, and  $V'_R = 5210$  meV for Cu(111), Cu(100), and Cu(110), respectively.

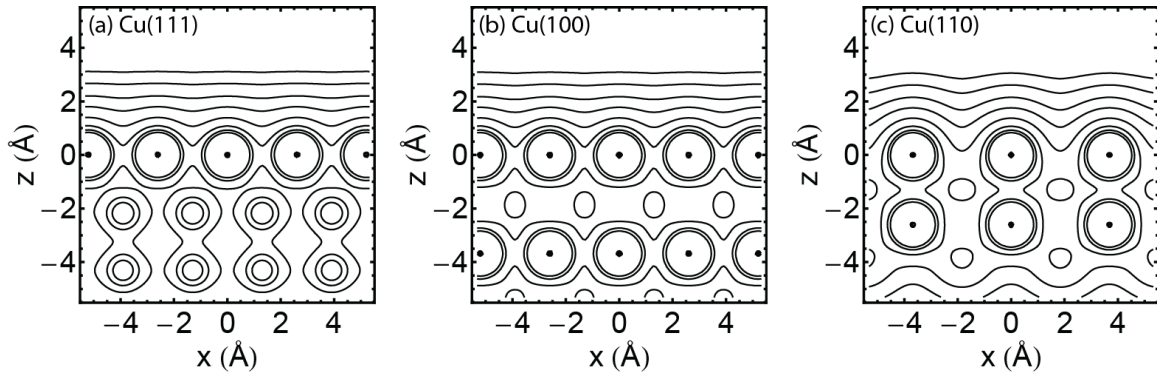


FIG. 1: Electron density profiles  $n_o(\mathbf{r})$  of the clean Cu(111), (100), and (110) surfaces calculated with the vdW-DF2 functional. The density contours take values in a nonlinear fashion. For easy comparison, the contour values in each plot are set to the same values: 0.1, 0.06, 0.03, 0.01, 0.003, 0.001, 0.0003, and 0.0001 in atomic units.

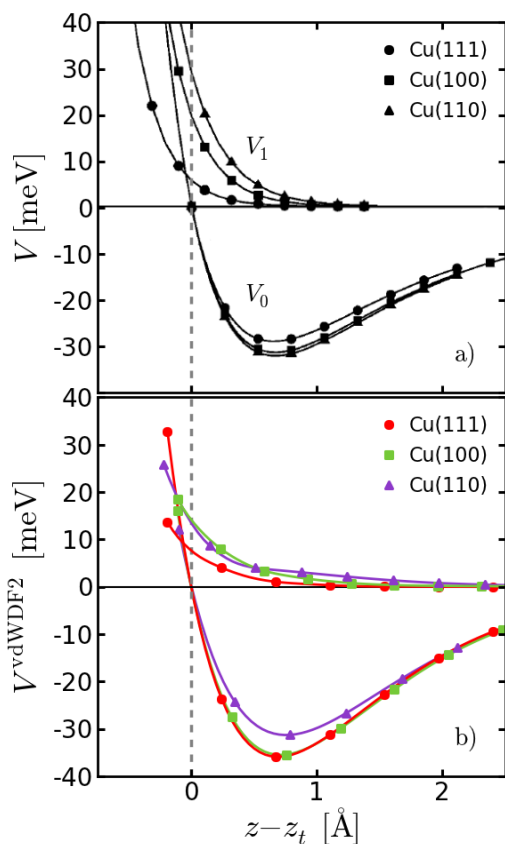


FIG. 2: (a) Physisorption interaction potentials,  $V$ , for  $\text{H}_2$  ( $\text{D}_2$ ) on Cu(111) (circles), Cu(100) (squares), and Cu(110) (triangles), in the form of lateral average,  $V_0(z)$  and corrugation  $V_1(z)$ . The potential functions  $V_0$  and  $V_1$  are defined in the text. The position  $z$  of the molecular center of mass is here given with respect to the classical turning point, i.e. the position  $z_t$  where a classical particle at energy  $\epsilon_i = 0$  would be reflected in the potential. Adapted from [15]. (b) The corresponding PECs calculated with vdW-DF2, averaged according to Eq. (7).

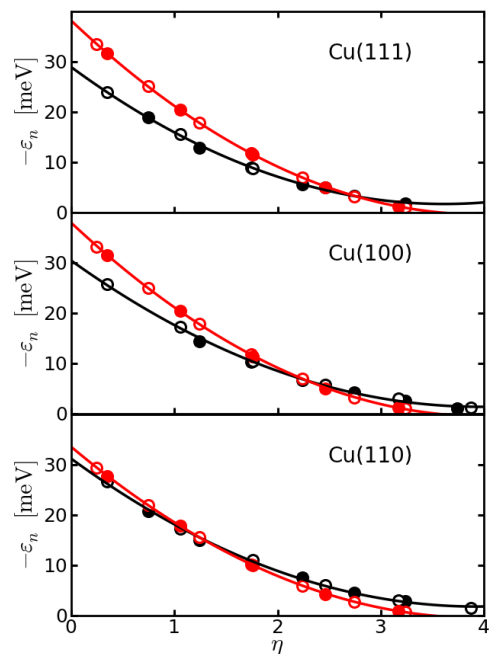


FIG. 3: Energy levels in the  $V_0$  potentials of Fig. 2(b) are plotted versus the mass-reduced level number  $\eta = (n + 1/2)/\sqrt{m}$ , where  $n$  is the quantum number and  $m$  the mass. Experimentally determined values are given by solid black and open black circles (the open circles being deuterium results). The theory results for the levels are identified by solid red and open red circles (the open circles being the deuterium results). Experimental values from [15].

of para- $\text{H}_2$  (p- $\text{H}_2$ ) and ortho- $\text{D}_2$  (o- $\text{D}_2$ ) and the normal species (n- $\text{H}_2$ , n- $\text{D}_2$ ) are thus available; this richness in data means that the data analysis is greatly simplified and the interpretation is clear. For instance, the rota-

tional anisotropy of the interaction has been determined via analysis of resonance structure resulting from the rotational ( $j, m$ ) sub-level splittings observed for n-H<sub>2</sub> and p-H<sub>2</sub> beams [36, 39]. Such knowledge permits a firm conclusion that the here-discussed measured bound-state energies,  $\epsilon_n$  (Fig. 3), refer to an isotropic distribution of the molecular orientation. The level assignment is compatible with a single gas-surface potential for the two hydrogen isotopes [16].

Figure 3 illustrates the analysis that leads to a single accurate gas-surface potential curve for each of the facets from the experimentally observed energies  $\epsilon_n$  [15–17]. The black open and filled circles represent measured  $\epsilon_n$  values. The procedure is an adaptation to surface physics of the Rydberg-Klein-Rees method of molecular physics [19]. The ordering is experimentally known and in this ordering, all  $\epsilon_n$  values fall accurately on a common curve [when plotted versus the mass-reduced level number  $\eta = (n + 1/2)/\sqrt{m}$ ]. The variation in the quantization levels reflects the asymptotic behavior of the potential curve and thus determines the value of  $C_{\text{vdW}}$  to a high accuracy and gives a good direct estimate of the well depth [16, 17]. A third-order polynomial fit to the data yields for  $\eta = 0$  a potential-well depth  $D = 29.5, 31.4, \text{ and } 32.3$  meV for the (111), (100), and (110) surfaces, respectively. This direct construction of an effective physisorption potential supplements the above-described experiment-based procedure [that instead uses the measured energies  $\epsilon_n$  to fit  $V_0(z)$  curves and obtain an even higher accuracy<sup>2</sup>].

Figure 2(a) also shows experiment-based determinations of the (rotationally averaged) amplitudes  $V_1(z)$  of the lateral corrugation. The measured intensities of the first-order diffraction beams provide (as described above) an estimate of the resulting lateral variation in the H<sub>2</sub>-Cu potential. The corrugation is very small,  $\sim 0.5$  meV at the potential well minimum. However, the existence of finite amplitudes  $V_1(z)$  is essential: The larger corrugation closer to the substrate contributes most importantly to the diffraction and resonance phenomena. In fact, it is a finite magnitude of  $V_1(z)$  that ensures a coupling to the in-plane crystal momentum and allows an elastic scattering event to satisfy the kinematical condition (5).

<sup>2</sup> We have tested the accuracy and consistency among the two experimental determinations of the effective physisorption potential for Cu(111). Specifically, we constructed an alternative  $V_0(z)$  form in which we directly inserted the Rydberg-Klein-Rees/Le Roy value of  $C_{\text{vdW}}$  and used the directly extracted well depth 31.4 meV [16] to specify an effective value of  $V_R'$ . We found that the lowest 4 physisorption eigenvalues of this potential then coincided within 3 percent of the measured  $\epsilon_n$  energies. In contrast, the experiment-based potentials  $V_0(z)$  described in Ref. [15] above and in potentials Fig. 2(a) reproduce the full set of measured energies (for all facets) to within 0.3 meV.

### III. VAN DER WAALS ACCOUNTS USED IN DFT CALCULATIONS

Noncovalent forces, such as hydrogen bonding and vdW interactions, are crucial for the formation, stability, and function of molecules and materials. In sparse matter the vdW forces are particularly relevant in regions with low electron density. For a long time, it has been possible to account for vdW interactions only by high-level quantum-chemical wave-function or Quantum Monte Carlo methods. The correct long-range interaction tail for separated molecules is absent from all popular local-density or gradient corrected XC functionals of density-functional theory, as well as from the Hartree-Fock approximation. Development of approximate DFT approaches that accurately model the London dispersion interactions [40, 41] is a very active field of research (reviewed in, for example Refs. 42–46).

To account for vdW interactions in computational physics traditional DFT codes are natural starting points. The vdW energy emanates from the correlated motion of electrons and there are proposals to account for it, like (i) DFT extended atom-pair potentials, (ii) explicit density functionals, and (iii) RPA in perturbation theory. Chemical accuracy is aimed for. Extensive physical and chemical systems are of great interest, including bio- and nanosystems, where the vdW interactions are indispensable. Describing bonds in a variety of systems with “chemical accuracy” requires that both strong and weak bonds are calculated. Strong covalent bonds are well described by traditional approximations, like the generalized gradient approximations (GGAs) [47–49], which are typically built into approaches (i) and (ii). vdW-relevant systems range from small organic molecules to large and complex systems, like sparse materials and protein-DNA complexes. They all have noncovalent bonds of significance. Quite a number of naturally and technologically relevant materials have already been successfully treated [46].

The methods (i) and (ii) are essentially cost free: speed is given by that of traditional DFT (for example GGA based). Such DFT calculations are competitive in terms of efficiency and broad applicability. Computational power is an important factor that is still relevant and an argument for choosing methods of types (i) and (ii) in many applications ahead of type (iii).

#### A. DFT extended by atom-pair potentials

A common remedy for the missing vdW interaction in GGA-based DFT consists of adding a pairwise interatomic  $C_6/R^6$  term ( $E_{\text{vdW}}$ ) to the DFT energy. Examples are DFT-D [50], TS-vdW [5], and alike [51, 52]. Refs. [53, 54] describe various other approaches also currently in use.

The DFT-D method is a popular way to add on dispersion corrections to traditional Kohn-Sham (KS) den-

sity functional theory. It is implemented into several code packages. Successively it has been refined to obtain higher accuracy, a broader range of applicability, and less empiricism. In the recent DFT-D3 version [4], the main new ingredients are atom-pairwise specific dispersion coefficients and cutoff radii that are both computed from first principles. The coefficients for new eighth-order dispersion terms are computed using established recursion relations. Geometry-dependent information is here included by employing the new concept of fractional coordination numbers. They are used to interpolate between dispersion coefficients of atoms in different chemical environments. The method only requires adjustment of two global parameters for each density functional, is asymptotically exact for a gas of weakly interacting neutral atoms, and easily allows the computation of atomic forces. As recommended by the authors of the method [4], three-body nonadditivity terms are not considered in our DFT-D3 study.

Another almost parameter-free<sup>3</sup> method for accounting for long-range vdW interactions from mean-field electronic structure calculations relies on the summation of interatomic  $C_6/R^6$  terms, derived from the electron density of a molecule or solid and reference data for the free atoms [5]. The mean absolute error in the  $C_6$  coefficients is 5.5%, when compared to accurate experimental values for 1225 intermolecular pairs, irrespective of the employed exchange-correlation functional. The effective atomic  $C_6$  coefficients have been shown to depend strongly on the bonding environment of an atom in a molecule [5].

## B. Explicit density functionals

Ground-state properties can be described by functionals of the electron density  $n(\mathbf{r})$  [1]. The functional  $E_{xc}[n(\mathbf{r})]$  for the XC energy is a central ingredient. The local-density approximation (LDA) [2, 55] and GGAs [47–49] do not describe the nonlocal correlations behind the vdW interactions. This subsection discusses explicit XC density functionals  $E_{xc}[n(\mathbf{r})]$ , focusing on the nonlocal correlation functional,  $E_c^{\text{nl}}[n(\mathbf{r})]$ .

In the vdW-DF, the vdW interactions and correlations are expressed in terms of the density  $n(\mathbf{r})$  as a truly nonlocal six-dimensional integral [6, 56, 57]. It originates in the adiabatic connection formula [58–60], and uses an approximate coupling-constant integration and an approximate dielectric function with a single-pole form. The dielectric function is fully nonlocal and satisfies known limits, sum rules, and invariances, has a pole strength

determined by a sum rule and is scaled to locally give the approximate gradient-corrected electron-gas ground-state energy. There are no empirical or fitted parameters, just references to general theoretical criteria.

Account for inhomogeneity is approximately achieved by a gradient correction, which is obtained from a relevant reference system. In the original vdW-DF version [6, 56, 57, 61], the slowly varying electron gas is used for this. The gradient correction is then taken from Ref. 62, as explained in the appendix of Ref. 56. Although promising results have been obtained for a variety of systems, including adsorption [46, 63], there is room for improvements. Recently another reference system has been proposed, with the argument that adsorption systems have electrons in separate molecule-like regions, with exponentially decaying tails in between. The vdW-DF2 functional uses the gradient coefficient of the B88 exchange functional [64] for the determination of the internal functional [Eq. (12) of Ref. 6] within the nonlocal correlation functional. This is based on application of the large- $N$  asymptote [65, 66] on appropriate molecular systems. Using this method, Elliott and Burke [67] have shown, from first principles, that the correct exchange gradient coefficient  $\beta$  for an isolated atom (monomer) is essentially identical to the B88 value, which had been previously determined empirically [64]. Thus in the internal functional, vdW-DF2 [3] replaces  $Z_{ab}$  in that equation with the value implied by the  $\beta$  of B88. This procedure defines the relationship between the kernels of vdW-DF and vdW-DF2 for the nonlocal correlation energy. Like vdW-DF, vdW-DF2 is a transferable functional based on physical principles and approximations. It has no empirical input.

The choices of exchange functional also differ. The original vdW-DF uses the revPBE [68] exchange functional, which is good at separations in typical vdW complexes [6, 56, 57]. At smaller separations [69–73], recent studies suggest that the PW86 exchange functional [74] most closely duplicates Hartree-Fock interaction energies both for atoms [70] and molecules [71]. The vdW-DF2 functional [3] employs the PW86R functional [71], which more closely reproduces the PW86 integral form at lower densities than those considered by the original PW86 authors.

## C. RPA

For first-principles electron-structure calculations, the random-phase approximation (RPA) to the correlation energy is presumably a suitable complement to the exact exchange energy [9]. The RPA to the correlation energy [75] incorporates a screened nonlocal exchange term and long-range dynamic correlation effects that underpin vdW bonding [9].

Hubbard, Pines and Nozieres [76] pointed out that RPA does have, at least, formal limitations in the description of local correlations (large momentum transfer).

<sup>3</sup> Both DFT-D and TS-vdW have a need to fix a cross-over function that is designed to minimize double counting of the semilocal correlation in regular DFT and the vdW contribution. The parameter of this cross over is often fitted to a reference system, for example S22.

This is because RPA treats same- and opposite-spin scattering on the same footing, thereby neglecting effects of Pauli exclusion in the description of the RPA correlation term. There exist several suggestions for RPA corrections [77]. A recent study [78] suggests a single-excitation extension for RPA calculations in inhomogeneous systems, thus lowering the mean average error for noncovalent systems [78].

Efficient RPA implementations have become increasingly available for solids [9, 79, 80] and molecular systems [81–84]. One implementation [9] gives the XC functional as

$$E_{xc} = E_{\text{EXX}} + E_c, \quad (6)$$

where the exact exchange energy  $E_{\text{EXX}}$  (Hartree-Fock energy) and the correlation energy  $E_c$ , given as the independent-particle response function, are all evaluated from KS orbitals by using for example plane-wave code and suitably optimized projector augmented wave (PAW) potentials that describe high energy scattering properties very accurately up to 100 eV above the vacuum level [85]. The operations scale like  $N^{6-7}$  and a high parallel efficiency can be reached [9].

#### D. Implementation aspects

The vdW-DF2 calculations are performed by using the ABINIT [86, 87] code with a plane-wave basis set and Troullier-Martins-type [88] norm-conserving pseudopotentials. The scalar-relativistic correction is included in the pseudopotentials for transition-metals. A kinetic energy cut-off of 70 Ry is used. For  $k$ -space integrations, a  $4 \times 4 \times 1$  Monkhorst-Pack mesh is used. For the partial occupation of metallic bands, we use the Fermi-Dirac smearing scheme with a 0.1 eV broadening width. With this setup the adsorption energies are converged within 1 meV. The vdW-DF total energy is calculated in a fully self-consistent way [56]. We adapted an implementation of the efficient vdW-DF algorithm [89] from SIESTA [90] for use within a modified version of ABINIT.

The surfaces are modeled by a slab of four atomic layers with a vacuum region of 20 Å in a periodic supercell. For the calculations on the (111), (100), and (110) surfaces we use the surface unit cells of  $3 \times 2\sqrt{3}$ ,  $3 \times 3$ , and  $2\sqrt{2} \times 3$ , respectively. Like benzene [91],  $\text{H}_2$  physisorbs on these metal surfaces, and the inertness associated with the high HOMO-LUMO gap hinders charge transfer at separations that are as long as typical vdW bonds. No dipole compensation is then needed: dipole-dipole interactions are negligible, as we have explicitly verified.

A thin slab should give sufficient accuracy, as long as there is no clear impact on the vdW-DF binding by the formation of surface states on Cu(111); otherwise, thicker slabs are in principle needed [92]. Here the focus is on a comparison across Cu facets. The previous Cu(111) study [18] used four layers. Our convergence test of the adsorption energy with respect to the slab thickness

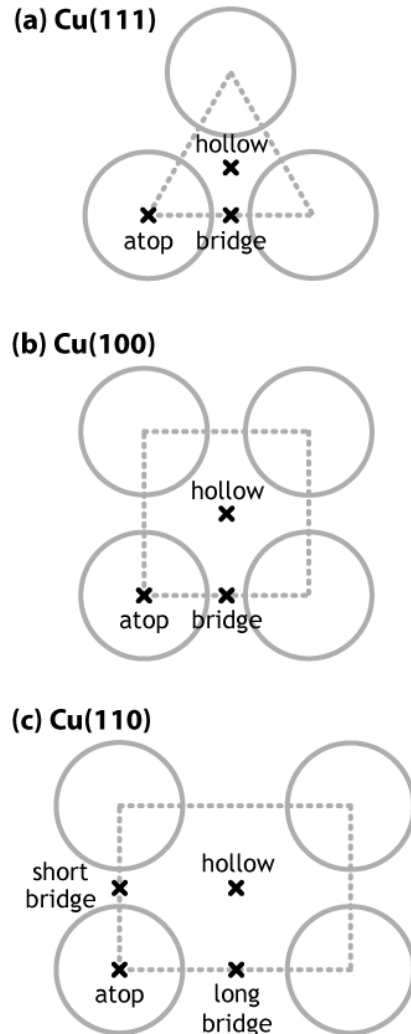


FIG. 4: High-symmetry positions on the low-index Cu surfaces. In our calculations the  $\text{H}_2$  molecule lies flat in one of these positions.

showed that a four-layer slab is thick enough to achieve convergence of the adsorption energies to within 0.5 meV (1%) when increasing the slab thickness by 50% for the Cu(110) and Cu(100) cases. We did not include a dipole correction, but our tests suggest a very limited effect:  $< 2\%$  and  $0.3\%$  deepening of the PEC for Cu(110) and Cu(100), respectively.

In the electron-structure calculations, the molecule is kept in a flat orientation above the high-symmetry positions or sites<sup>4</sup> on the Cu surfaces, as indicated in Figure 4. Some test points indicate that the total energy depends

<sup>4</sup> In the description of the DFT calculations we refer to these positions as “sites” but note that there is a large zero-point motion perpendicular to the surface and that the experimentally relevant molecules move with a large in-surface kinetic energy while trapped in the physisorption well.



very little on orientation. For instance, in a “worst-case” situation, the energy change by a 90-degree in-plane rotation of  $\text{H}_2$  at the long bridge site of  $\text{Cu}(110)$  is 0.92 meV, in a fixed-height rotation. If  $\text{H}_2$  is moved to the equilibrium adsorption height, which is 0.04 Å lower, the energy difference increases to 0.97 meV. This variation is much smaller than the lateral corrugation ( $\sim 4$  meV in this facet) and out-of-plane rotation ( $\sim 5$  meV).

To estimate the magnitude of the error introduced by neglecting the angular average (that automatically is included in the experimental data) we perform a separate calculation of  $\text{H}_2$  in an upright position on the  $\text{Cu}(111)$  surface using the optimal  $\text{H}_2$ -to-surface separation. This calculation results in a downward energy shift of 4.8 meV, which is the amplitude of the angular variation at the optimal separation. However, for an isotropic  $\text{H}_2$  wave function this shift corresponds roughly to lowering of the ground state energy by merely  $1/3 \times 4.8$  meV = 1.6 meV, assuming a simple sinusoidal energy variation in the angular space. The energy of higher-order states would also be lowered, but to a lesser extent. This estimate thus indicates that the effect of the angular energy dependence is merely a minor quantitative correction to the eigenvalues.

The PECs are calculated with vdW-DF2 for the high-symmetry sites, and from these the laterally averaged potential  $V_0$  is approximately obtained. In turn, the bound quantum states in the  $V_0$  potential well are calculated by solving the corresponding Schrödinger equation.

The theoretical values are generated for a number of discrete points, surface positions (atop, hollow, long bridge, and short bridge), and separations  $z$ , while experimental data (Fig. 2) are presented as lateral averages and functions of separation  $z$ . To connect the two, some approximation has to be made to extract the laterally averaged result out of the discrete one.

To capture the effects of the surface topology and to use the surface-lattice points used in the DFT calculations, we find the following approximate average reasonable:

$$V_0 = \frac{1}{4}(V_{\text{hollow}} + V_{\text{atop}} + V_{\text{long-bridge}} + V_{\text{short-bridge}}) \quad (7)$$

where  $V_{\text{long-bridge}} = V_{\text{short-bridge}}$  for the (100) and (111) surfaces. One argument in favor of this approximation is the fact that the (100) surface has twice as many bridge sites as there are atop sites. Approximation (7) is used in our comparisons between our vdW-DF2-determined potential  $V_0$  and the experimentally determined one (Fig. 2) and in our generation of eigenvalues used to relate to the experimental ones [which are defined in terms of a laterally averaged and  $\text{H}_2$  rotational-angle averaged potential  $V_0(z)$ ] (Fig. 3). Reference 18 uses the atop PEC on  $\text{Cu}(111)$ , which is an adequate choice due to the small corrugation of the  $\text{Cu}(111)$  surface.

For the discussion of the relation between corrugation and  $V_1$ , the classical turning point is the relevant separation. The corrugation of the PEC minimum is smaller,

but still representative of the expected variation in the probability for the  $\text{H}_2$  trapping.

The original report of experimental results also covers a potential  $V_2(z)$ . It represents the min-to-max variation of the lateral average of the rotational anisotropy. A full appreciation of the accuracy of the comparison between the experimentally determined  $V_0(z)$  and  $V_0^{\text{vdW-DF2}}(z)$ , based on existing vdW-DF2 calculations, would benefit from an understanding of  $V_2(z)$ . To indicate that this is unlikely to have any large consequence for the three Cu surfaces, a simple estimate of the rotation angle effect is made above. This should be considered as a stimulus for further refinement of the testing.

## IV. RESULTS AND BENCHMARKING

Here we apply a benchmark from surface physics, whose virtues have challenged theory for a long time. Data are provided for (i) energy eigenvalues,  $\epsilon_n$ , for  $\text{H}_2$  and  $\text{D}_2$  in the PEC well, which have direct ties to measured reflection intensity, (ii) the laterally averaged physisorption potential,  $V_0$ , which is derived from measured data, the extracted PEC, and (iii) the corrugation,  $V_1$ , also derived from measured data.

### A. Benchmarking strategies

Evaluation of XC functionals is often made by comparing with other theoretical results in a systematic way, for instance, the common comparison with S22 data set [12, 13, 93–95]. These sets have twenty-two prototypical small molecular dimers for non-covalent interactions (hydrogen-bonded, dispersion-dominated, and mixed) in biological molecules. They provide PECs at a very accurate level of wave-function methods, in particular the CCSD(T) method. However, by necessity, the electron systems in such sets are finite in size. The original vdW-DF performs well on the S22 data set, except for hydrogen-bonded duplexes (underbinding by about 15% [3, 46]). Use of the vdW-DF2 functional reduces the mean absolute deviations of binding energy and equilibrium separation significantly [3]. Shapes of PECs away from the equilibrium separation are greatly improved. The long-range part of the vdW interaction, particularly crucial for extended systems, has a weaker attraction in the vdW-DF2, thus reducing the error to 8 meV at separations 1 Å away from equilibrium [3].

Recently, other numbers for the S22 benchmark on vdW-DF2 have been published [96]. The two calculations differ in the treatment of the intermolecular separation, one relaxed [3] and the other unrelaxed [96]. Of course, absence of relaxations does lead to an appearance of worse performance.

Experimental information provides the ultimate basis for assessing functionals. The vdW-DF functional

has been promising in applications to a variety of systems [46], but primarily vdW-bonded ones. Typically, the calculated results are tested on measured binding-energy and/or bond-length values that happen to be available. The vdW-DF2 functional has also been successfully applied to some extended systems, like graphene and graphite [3], metal-organic-frameworks systems [97], molecular crystal systems [98], physisorption systems [99, 100], liquid water [101] and layered oxides [102]. However, the studies are of the common kind that focus on comparison against just a few accessible observations.

Accurate experimental values for the eigenenergies of  $H_2$  and  $D_2$  molecules bound to Cu surfaces [15, 16] motivate theoretical account and assessment. This knowledge base covers results for the whole shape of the physisorption potentials. Here calculations on several Cu facets allow studies of trends and a deeper analysis. The extensive report of vdW-DF2 results in Figure 5 serves as a starting point.

### B. PECs from vdW-DF2

PECs are calculated for  $H_2$  in atop, bridge, and hollow sites on the Cu(111), (100), and (110) surfaces. The resulting benchmark for vdW-DF2 is also compared (below) with those of two other vdW approximations, the DFT-D3 [4] and TS-vdW [5] methods.

To emphasize various aspects of the PECs and make valuable use of the numerical accuracy, the next few sections (and figures) highlight various aspects of the vdW-DF2 results. For each position  $z$  of the molecular center of mass, we compare the averaged potential functions  $V_0(z)$  and  $V_1(z)$ , in Fig. 2, and find strong qualitative agreement. For instance, at the potential minimum of  $V_0(z)$ , Figure 2 gives for  $V_1(z)$  the approximate values 1, 3, and 4 meV for (111), (100), and (110), respectively, in quantitative agreement with the illustration of theoretical results analysis. The insensitivity to facets and the corrugation is discussed in greater detail below.

### C. Isotropy of lateral averaged potentials

An important feature of the experimental  $V_0(z)$  in Figure 2 is the similar sizes of the well depths (range 29–32 meV) and separations (around 3.5 Å) on the (111), (100), and (110) surfaces. This isotropy, i.e., similarity of physisorption potential among different facets, is interesting and perhaps surprising because Cu(111) contains a metallic surface state, whereas Cu(100) and Cu(110) do not.

The most striking feature of the vdW-DF2 results for the  $H_2$ -Cu PEC is probably that it is able to reproduce this isotropy. From Figure 5 we find physisorption depths in the interval 35–39 meV and separations in the range 3.3–3.6 Å. From the experimentally more relevant later-

ally averaged  $V^{\text{vdW-DF2}}(z)$ , Fig. 2(b), we find physisorption depths 31–36 meV.

The isotropy is emphasized in Figure 6 by plotting the vdW-DF2 PECs on the atop sites of each surface. The curves lie very close to each other, both in the Pauli-repulsion region at short separations, dominated by  $V_R$  in the traditional theory [Eqs. (1)-(3)], and in the vdW-attraction region at large separations. They differ only discernibly in the  $V_R$ -region, which can be understood in terms of the higher electron density,  $n_o(\mathbf{r})$  on the atop site on the dense (111) surface.

Both agreements and differences are found in the trends (with facets) of the experiment-based [ $V_0(z)$ ,  $V_1(z)$ ] and vdW-DF2 based [ $V^{\text{vdW-DF2}}(z)$ ] characterizations, Figure 2(a) and (b). vdW-DF2 reproduces the trend in ordering and roughly the magnitudes in the modulation amplitudes [ $V_1(z)$ ] at the classical turning points (identified as position ‘0’ in Figure 2). As shown in Fig. 6, vdW-DF2 also reproduces the ordering of separations corresponding to physisorption minima, correctly decreasing as (111) > (100) > (110). On the other hand, for  $V_0(z)$  the physisorption depth varies as (111) > (100) > (110) whereas in  $V^{\text{vdW-DF2}}(z)$  the depth varies (110) > (100) > (111). The largest relative difference, 25%, is found for Cu(111). The set of physisorption depths on the three facets are reproduced with an average confidence of 15%.

It is interesting that it is the facet Cu(111), containing a metallic surface state, that breaks the trend and presents the largest deviation from experiment [18]. The surface state accounts for 25% of the static screening arising from an external field [92], and can certainly change the electrodynamical response that defines the vdW bonding. However, the surface-state formation does suggest that there must be a depletion in the bonding contribution from the bulk-like states. It is not a priori clear what the net effect will be at binding distances. These aspects are exciting but require a separate future investigation, and are beyond the scope of the present paper.

### D. Corrugation

Another striking feature of the PECs is the variation with the density corrugation of each surface. In Figure 1, the density profiles of the clean Cu(111), (100), and (110) surfaces indicate how the corrugation may vary. For fcc metals the (111) surface is the most dense, while the (100) and (110) surfaces are successively more open and thus corrugated. The trend is reflected in the PECs. These clear effects on the PEC are illustrated by the calculated PECs for  $H_2$  in atop, bridge, and hollow sites on the Cu(111), (100), and (110) surfaces (Fig. 5). From being small on the flat and dense (111) surface [Fig. 5(a)], the corrugation grows from (111) to (100) and from (100) to (110), just as expected from the above reasoning.

Figure 7 shows the variations in the adsorption ener-

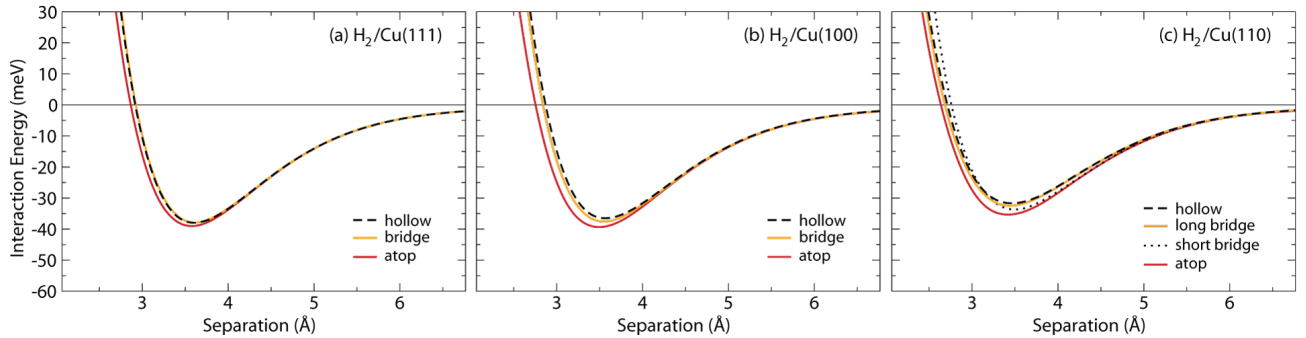


FIG. 5: Calculated PECs for  $H_2$  in atop, bridge, and hollow sites on the (a) Cu(111), (b) (100), and (c) (110) surfaces, calculated with the vdW-DF2 functional.

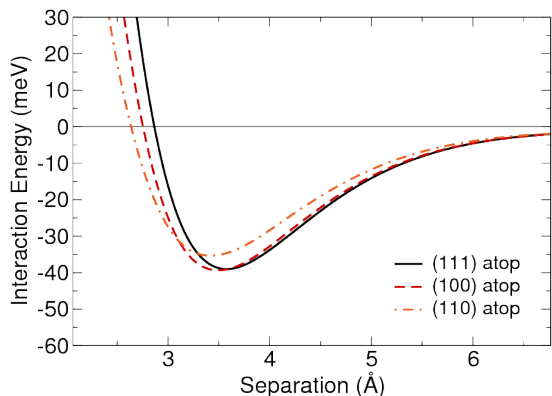


FIG. 6: PECs of  $H_2$  on atop sites of the Cu(111), (100), and (110) surfaces calculated with the vdW-DF2 functional.

gies relative to the value of the atop configuration on the various facets. We choose to report the corrugation at the PEC minimum. The corrugation at the classical turning point is likely a stronger indicator of the strength of the elastic scattering that traps the incoming  $H_2$  molecules, Sec. II.

On all three facets, the calculated stable site is atop (Figs. 5 and 7). This result can be understood with a simple argument based on the traditional model and a tight-binding (TB) description of the electrons. Equations (1)–(3) separate the potential energy into repulsive and attractive parts,  $V_R$  and  $V_{vdW}$ , respectively. Close to the minimum point, the vdW attraction,  $V_{vdW}$  (Eq. (3)) gets stronger in the direction towards the surface. The repulsion terms  $V_R$  prevent the admolecule from benefiting from this by going even closer. Equation (4) reflects that higher density gives higher repulsion, and the density profiles in Figure 1 show the proper order. For a more general discussion, see Ref. [103].

While the electron density (Fig. 1) is characterized by only one kind of corrugation, the corrugation of the PECs depends on where the probe hits the PEC. From Figure 5 we can envisage different corrugation values for different  $z$  values. For the reflection-diffraction experiment one

can argue that  $H_2$  molecules coming in to the surface with a positive kinetic energy, i.e., at or above the energy of the classical turning point, are particularly relevant.

### E. Corrugation and exchange functional

To clarify the underlying cause of corrugation, a different adsorption system is first studied. Benzene on the Cu(111) surface is known to be a true vdW system [91]. Interaction-energy values for the benzene molecule at various positions on the Cu(111) are calculated with five different density functionals, all accounting for vdW forces, and shown in Figure 8. The functionals differ by the differing strengths of vdW attraction in vdW-DF and vdW-DF2, but in particular by different exchange approximations, revPBE [6, 68], C09 [73], optPBE [72], and PW86R [3, 71, 74]. While the binding-energy value is not so sensitive to choice of functional, corrugation-energies values are.

Therefore, the accurate results of a reflection-diffraction experiment are valuable in several respects. On one hand, they support that the traditional model is right in its separation in Eqs. (1)–(3) and their attaching the repulsive wall,  $V_R$ , to Pauli repulsion [17] and thus to exchange. On another hand, they are able to discriminate between different approximations for the exchange functional. Similar results have also been calculated for benzene on graphene [104].

### F. Comparison with experiment-related quantities

Comparison of the calculated results on the  $H_2$ -Cu systems with the model used for the analysis of the experimental data [15, 16] is next done for the laterally averaged potential  $V_0(z)$ , the potential derived from experiment. The experiment-derived results in Figure 2 are redrawn in Figure 9 as the experimental physisorption potential for  $H_2$  on Cu(111). Figure 9(a) shows our comparison of PECs calculated using vdW-DF and vdW-DF2, respectively, drawn against the experimental physi-

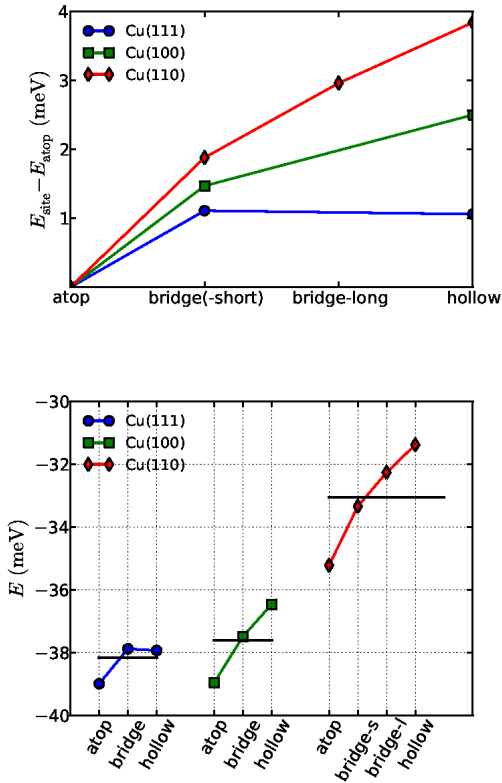


FIG. 7: Corrugation for H<sub>2</sub> on Cu(111), Cu(100), and Cu(110) (a) illustrated by the lateral variation of the calculated adsorption energy of H<sub>2</sub> at each high-symmetry position on these surfaces; (b) illustrated by calculated adsorption-energy values of H<sub>2</sub> on these surfaces. The black horizontal lines indicate approximate site averages, analogous to Eq. (7). Subtracting the atop value from each average gives 1.3, 1.6, and 2.5 meV as lateral variations on the (111), (100) and (110) surfaces, these numbers can be interpreted as measures of the corrugation.

sorption potential for H<sub>2</sub>, originally published in Ref. 18. The Cu(111) surface is chosen for its flatness that gives clarity in the analysis and eliminates several side-issues that could have made interpretations fuzzier. Several qualitative similarities are found for both vdW-DF and vdW-DF2 functionals. The vdW-DF2 functional gives PECs in a useful quantitative agreement with the experimental physisorption curve, for instance with respect to well depth, equilibrium separation, and curvature of PEC near the well bottom, and thus zero-point vibration frequency. Comparisons of full PECs are also parts of the benchmarking.

### G. Other methods

Well depths and equilibrium separations for H<sub>2</sub> on Cu(111) are seen as PEC minimum points in Figure 9(a).

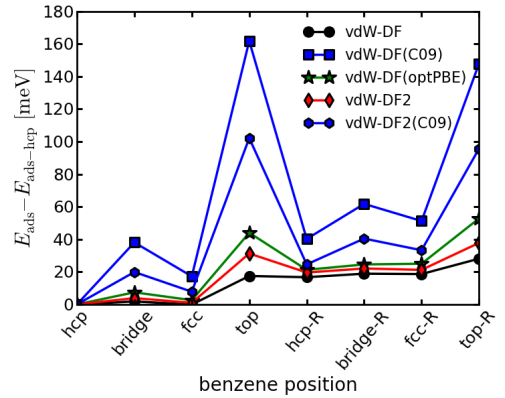


FIG. 8: Interaction-energy values for benzene at various positions on the Cu(111) surface, calculated with different XC functionals. It shows that corrugation energies are sensitive to choice of functional, according to calculations with vdW-DF [6], vdW-DF(C09) [73], vdW-DF(optPBE) [72], vdW-DF2 [3], and vdW-DF2(C09).

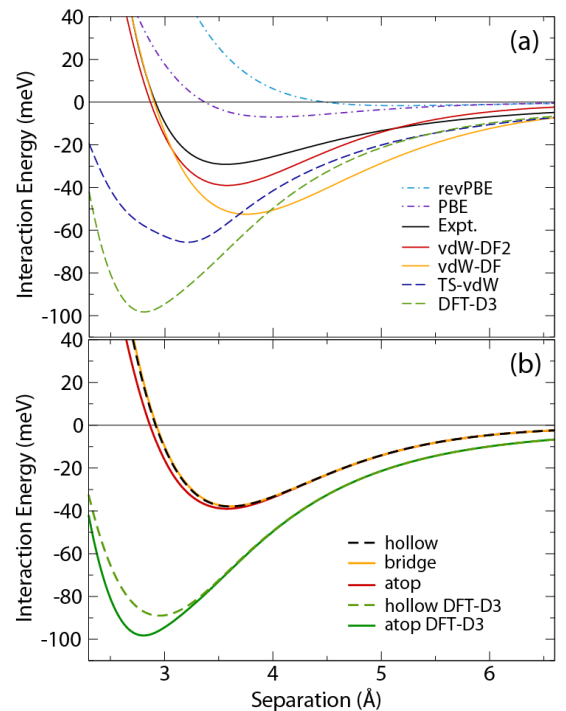


FIG. 9: (a) Experimentally determined effective physisorption potential for H<sub>2</sub> at atop site on the Cu(111) surface [15], compared with potential-energy curves for H<sub>2</sub> on the Cu(111) surface, calculated for the atop site in GGA-revPBE, GGA-PBE, vdW-DF2, vdW-DF [18], TS-vdW [5], and DFT-D3(PBE) [4]. Partly adapted from Ref. 18. (b) Comparison of PECs for atop and hollow sites on Cu(111) calculated with vdW-DF2 and DFT-D3.

As the corrugation is so small, it should suffice to use only the atop result. More precise value pairs [18] are  $(-28.9 \text{ meV}; 3.5 \text{ \AA})$  for  $V_0(z)$  extracted from experiment [15],  $(-53 \text{ meV}; 3.8 \text{ \AA})$  as calculated with the vdW-DF functional,  $(-39 \text{ meV}; 3.6 \text{ \AA})$  with the vdW-DF2 functional,  $(-98 \text{ meV}; 2.8 \text{ \AA})$  with the DFT-D3(PBE) method [4], and  $(-66 \text{ meV}; 3.2 \text{ \AA})$  with the TS-vdW method [5]. The striking discrepancy between the three major types of accounts for vdW in extended media is discussed below.

As reported for example in Ref. 18, the LDA and GGA functionals do not describe the nonlocal correlation effects that give vdW forces. They also misrepresent the PECs. The minima are too shallow and the equilibrium separations are too large [61].

In Figure 9, the DFT-D3(PBE) curves are the results of calculations with the DFT-D3 corrections [4] added on top of the PBE PECs. Figure 9(b) compares a DFT-D3 PEC at the hollow site of the (111) surface with that of an atop site [same as the one in Fig. 9(a)] [18]. The energy difference between atop and hollow adsorption-energy values is found to be 11 meV. If this corrugation at  $z_{\min}$  were a measure of the corrugation, the corrugation by DFT-D3 on Cu(111) would be 11 meV, or 11 times as large as that given by vdW-DF2 (1 meV).

It is interesting to note that the TS-vdW method delivers a PEC [Fig. 9(a)] similar to that from DFT-D3 [Fig. 9(a) and (b)], although not quite as deep.

## H. Energy levels

The quantum-mechanical motion of the  $\text{H}_2$  molecule in the various (laterally averaged) potentials (Fig. 2) can be calculated and for the motion perpendicular to the surface be described by, e.g., the bound-state eigenenergy values. Figure 3 presents and compares results from experiment and theory. The experimental curve, identified by filled ( $\text{H}_2$ ) and empty ( $\text{D}_2$ ) black circles may be analyzed [15–17] within the traditional theoretical picture [25, 26] of the interaction between inert adsorbates and metal surfaces, Section II. The experimental level sequence in Figure 3 can be accurately reproduced ( $< 0.3 \text{ meV}$ ) by such a physisorption potential [16, 17] (Fig. 2), having a well depth of 28.9 meV and a potential minimum located 3.5  $\text{\AA}$  outside the topmost layer of copper ion cores. From the measured intensities of the first-order diffraction beams, a very small lateral variation of the  $\text{H}_2$ -Cu(111) potential can be deduced,  $\sim 0.5 \text{ meV}$  at the potential-well minimum.

The vdW-DF2 theory results for the levels are identified by filled ( $\text{H}_2$ ) and empty ( $\text{D}_2$ ) red circles. The theoretical results are constructed from the calculated vdW-DF2 PECs in Figure 5 by first providing an estimate for the laterally averaged potential  $V_0^{\text{vdW-DF2}}(z)$  for each facet, according to Eq. (7).

We note that unlike the experimental results [which define  $V_0(z)$ ], the variation in  $V_0^{\text{vdW-DF2}}(z)$  does not reflect an average over the angles of the  $\text{H}_2$ . We also note

that this is a small effect, Section III D.

Figure 3 documents good agreement in results from vdW-DF2 and experiment for the energy levels on each of the Cu facets. The eigenvalues have the same order as in the experimental results (Fig. 3), indicating good agreement between the calculated and measured average potentials.

There are some discrepancies between the eigenvalues for  $\text{H}_2$ . This signals that the vdW-DF2 functional might not give the right shape for the PEC of  $\text{H}_2$  on Cu(111). The vdW-DF2 PEC is judged to lie close to the experimental physisorption potential, both at the equilibrium position and at separations further away from the surface, and is thus described as “promising” [18]. The same applies for  $\text{H}_2$  on Cu(100) and Cu(110).

## I. Summary

Having access to the full PEC, including the shape of the potential and asymptotic behavior, allows a more stringent assessment of the theoretical results. This is in addition to the many other conclusions that Figure 9 gives.

*PECs from vdW-DF2:* The general picture for  $\text{H}_2$  on Cu(111) of Ref. 18 applies also to the Cu(100) and (110) surfaces. Going from the densest surface, Cu(111), to the more open surfaces, the changes are small. The vdW-DF2 description is good to 25% in calculating the potential depth in the worst case, Cu(111). vdW-DF2 describes the mean physisorption well depths (averaged over all three facets) to within 15% of the experimental results.

*Lateral average:* Approximate lateral averages of the PECs,  $V_0$ , have a fair agreement with those derived from experiment (Fig. 2).

*Isotropy:* On the fcc metal Cu, the PECs of  $\text{H}_2$  are almost independent of facet orientation (Figs. 5 and 6).

*Corrugation:* On each surface, the PECs vary (Figs. 5–8) with the density corrugation (Fig. 1). For fcc metals the (111) surface is most dense, and (100) and (110) are successively more open and corrugated. For the calculated PECs for  $\text{H}_2$  in atop, bridge, and hollow sites on the Cu, trends and magnitudes of  $V_1$  agree with experimental findings (Figs. 2 and 7).

*Corrugation and exchange functional:* By the example of the benzene molecule on the Cu(111) surface, calculations with several different density functionals that account for vdW forces show that corrugation-energies values are sensitive to functionals that differ by different exchange approximations. Therefore, the accurate results of a reflection-diffraction experiment are valuable for discriminating between exchange functionals (Fig. 8). The vdW-DF2 functional uses a good exchange approximation.

*Comparison with experiment-related quantities:* The experiment-derived results are shown in Figures 2 and 9 as the experimental physisorption potential for  $\text{H}_2$  on

Cu(111). Comparison of PECs calculated with vdW-DF and vdW-DF2 show that the vdW-DF2 functional gives PECs in reasonable agreement with the experimental physisorption curve. It should be emphasized that the experimental physisorption curve involves an average over all sites.

*Other functionals:* PECs calculated with several different methods for H<sub>2</sub> on Cu(111) in atop and hollow positions show a striking discrepancy between the results from the DFT-D3 [4] and TS-vdW [5] methods on the one hand, and those of vdW-DF2 and experiment on the other hand. This discrepancy is traced back to the fact that pair potentials center the interactions on the nuclei and do not fully reflect that important binding contributions arise in the wave function tails outside the surface.

*Energy levels:* The energy levels in the H<sub>2</sub>-Cu PEC wells (Fig. 3) are calculated for all facets and compared with the experimental ones (Fig. 3). Agreement with experimental results is gratifying.

We judge the performance of vdW-DF2 as very promising. In making this assessment we observe that (i) vdW-DF2 is a first-principles method, where characteristic electron energies are typically in the eV range, and (ii) the test system and results are very demanding. The second point is made evident by the fact that other popular methods deviate significantly more from the experimental curve. For instance, application of the DFT-D3(PBE) method [4] (with atom-pairwise specific dispersion coefficients and cutoff radii computed from first principles) gives (−98 meV; 2.8 Å) for the PEC minimum point. The good agreement of the minima of the vdW-DF2 and experimental curves are encouraging, and so is the relative closeness of experimental and calculated eigenenergy values in Figure 3. The discrepancies between the eigenvalues signal that the vdW-DF2 PEC might not express the exact shape of the physisorption potential for H<sub>2</sub> on Cu(111).

## V. COMPARISONS AND ANALYSIS

Figures 9(a) and (b) show that the vdW-DF2 functional and DFT-D3 and TS-vdW methods give very different results. We could have made this point even stronger by showing also results for corrugation and energy values. However, we believe that comparisons of the PECs at atop and hollow sites on the Cu(111) surface suffice. No doubt, H<sub>2</sub> on Cu is a demanding case for all methods. The local probe H<sub>2</sub> on Cu avoids smearing-out effects, unlike for example graphene and PAHs, and incipient covalency, unlike H<sub>2</sub>O and CO, and is thus a pure vdW system and a lateral-sensitive one.

We trace the differing of the results with the vdW-DF2 and DFT-D3 methods to the differences in the descriptions of the vdW forces. The vdW-DF2 and similar functionals describe interactions between all electrons, while DFT-D3 and TS-vdW methods rely on atom-pair interactions. Even if large efforts are put into mimicking

the real electron-charge distribution by electron-charge clouds around each atom nucleus, this has to be a misrepresentation of surface-induced redistributions of electronic charge in a general-geometry correction of a traditional DFT calculation. There is no mechanism for Zaremba-Kohn effects.

The Zaremba-Kohn formulation of physisorption is not built in explicitly into the vdW-DF2 functional. However, the interactions of the electrons are built in into the supporting formalism in the similar way as in the derivation of the Zaremba-Kohn formula.

## VI. CONCLUSIONS

Accurate and extensive experimental data are used to benchmark calculational schemes for sparse matter, that is, methods that account for vdW forces. Reflection-diffraction experiments on light particles on well-characterized surfaces provide accurate data banks of experimental physisorption information, which challenge any such scheme to produce relevant physisorption PECs. PECs of H<sub>2</sub> on the Cu(111), (100), and (110) surfaces are here studied. Accuracy is high even by the surface-physics standards and is here provided thanks to diffraction kinematical conditions giving sharp resonances in diffraction beam intensities. We propose that such surface-related PEC benchmarking should find a broader usage.

The vdW-DF, vdW-DF2, DFT-D3, and TS-vdW schemes are used, and results are compared. The first two are expressions of the vdW-DF method, that is, nonempirical nonlocal functionals in which the electrodynamic couplings of the plasmon response produces fully distributed contributions to vdW interactions; Like in the Zaremba-Kohn picture [25], they permit the extended conduction electrons to respond also in the density tails outside a surface. The latter two are examples of DFT extended with vdW pair potentials and represent the dispersive interaction through an effective response and pair potentials located on the nuclei positions. Several qualitative similarities are found between the vdW-DF and vdW-DF2 functionals. The vdW-DF2 functional gives PECs in a useful qualitative and quantitative agreement with the experimental PECs. This is looked at for well depths, equilibrium separations, and curvatures of PEC near the well bottom, and thus molecular zero-point vibration frequency. The DFT-D3 and TS-vdW schemes give PEC results that deviate significantly more from experimental PECs. The benchmark with the experimental H<sub>2</sub>/Cu scattering data is thus able to discriminate between the results of pair-potential-extended DFT methods and vdW-DF2. The differences suggest that it is important to reflect the actual, distributed location of the fluctuations (plasmons) that give rise to vdW forces.

The vdW-DF2 density functional benchmarks very well against the S22 data sets [3]. It is also the functional being more extensively compared between experi-

ment and theory here. Certain very well-pronounced features, like isotropy of the H<sub>2</sub>-Cu PEC, the (111), (100), and (110) PECs being close to identical, and the clear trend in its corrugation that grows in order (111) < (100) < (110), are well described. The calculated  $V_1$  results are found to be close to the experimental ones, thereby being almost decisive on exchange functionals. The energy levels for the quantum-mechanical motion in the H<sub>2</sub>-Cu PEC agree in a gratifying way.

The accuracy of this experiment is also shown to be valuable for discriminating between exchange functionals (Fig. 6). The vdW-DF2 functional is found to apply a good exchange approximation.

The vdW-DF2 is found promising for applications at short and intermediate distances, as is relevant for adsorption. However, the accuracy of experimental data is high enough to stimulate a more detailed analysis of all aspects of the theoretical description. This should be valuable for further XC-functional development. For instance, some discrepancies are found for the eigenvalues. They signal that the vdW-DF2 PEC for H<sub>2</sub> on Cu(111) might not have a perfect shape. Additional physical effects could be searched for. The metallic sur-

face state on Cu(111) might be one source; It is possible that the metallic nature of the H<sub>2</sub>/Cu(111), H<sub>2</sub>/Cu(100), and H<sub>2</sub>/Cu(110) systems motivates modifications in the description of the electro-dynamical response inside the nonlocal functional. For a well established conclusion, a more accurate theory is called for.

In any case, H<sub>2</sub>/Cu physisorption constitutes possibilities for benchmarking theory descriptions and represents a very strong challenge for the density functional development.

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